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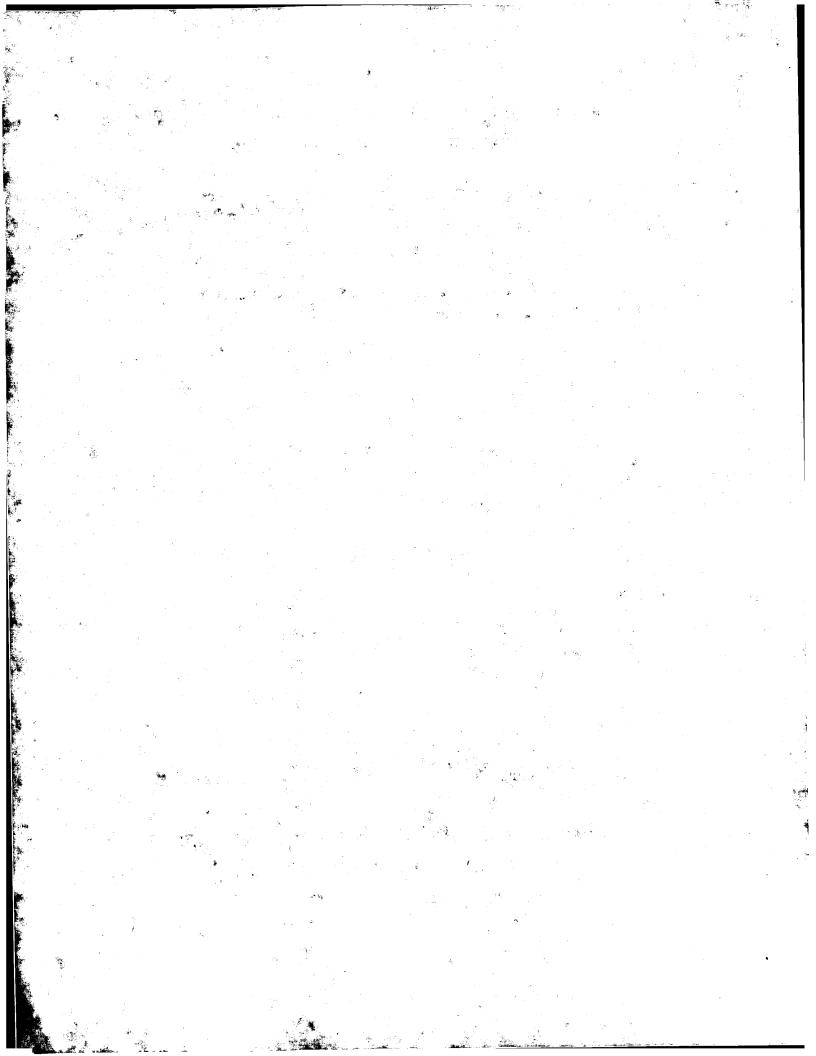
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(11) Publication number:

0 151 346

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#### **EUROPEAN PATENT APPLICATION**

(21) Application number: 84308760.2

(5) Int. Cl.4: **C** 03 **C** 3/089 C 03 C 3/091, C 03 C 4/08

22 Date of filing: 14.12.84

(30) Priority: 13.01.84 FR 8400481

Date of publication of application: 14.08.85 Bulletin 85/33

(84) Designated Contracting States: BE DE GB IT NL

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(57) This invention is concerned with the production of glasses suitable for ophthalmic applications having refractive indices of 1.523 ± 0.004, Abbe number between 51-59, densities less than 2.43g/cm³, transmissions at 400nm in 2 mm thickness greater than 89%, and a UV cutoff between 310-335nm, said glasses consisting essentially, in weight percent on the oxide basis, of:

| SiO,                           | 49-71 | $Li_2O + Na_2O + K_2O$   | 8-20  |
|--------------------------------|-------|--|-------|
| B <sub>2</sub> O <sub>3</sub>  | 5-26  | TiO <sub>2</sub>   | 1.8-6 |
| Al <sub>2</sub> O <sub>3</sub> | 0-14  | ZrO <sub>2</sub>   | 0-5.5 |
| Li <sub>2</sub> O              | 0-4   | As <sub>2</sub> O <sub>3</sub> and/or Sb <sub>2</sub> O <sub>3</sub> | 0-0.7 |
| Na <sub>2</sub> O              | 0-16  | Cl and/or Br   | 0-1   |
| K <sub>2</sub> O               | 0-20  | CaO and/or MgO   |       |
| 1124                           |       | and/or ZnO and/or BaO  | 0-4   |

<sup>(54)</sup> Glasses for ophthalmic applications.

## GLASSES FOR OPHTHALMIC APPLICATIONS

The present invention relates to glasses for ophthalmic applications which, in addition to the properties of ultraviolet radiation (UV) absorption and high transmission in the visible (absence of coloration), have a low density, as well as corrective lenses produced from these glasses.

These glasses are characterized by an index of refraction (n<sub>d</sub>) of 1.523 ± 0.004, an Abbe number (V<sub>d</sub>) between 51-59, and a density (D) less than 2.43g/cm<sup>3</sup>. Their transmission at 400nm is, for a thickness of 2mm, greater than 89%. Their UV cutoff, defined as the wavelength for which the transmission is equal to 1% for a thickness of 2mm, is between 310-335nm. Moreover, they exhibit excellent chemical durability (A.O. test).

Lenses of inorganic glass ("white" or tinted) for ophthalmic use have, to a very great extent, an index of refraction  $n_{\rm d}=1.523$ . The "white" (or tinted) glasses used at the present time and which exhibit in certain cases a UV cutoff higher than 300nm, have a density at least equal to  $2.54 {\rm g/cm}^3$ . A reduction of the density and, consequently, of the weight of the lens, offers an obvious advantage for the wearer and

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that regardless of the power of the lens. The lightening of the lens by the glasses of this invention is about 6.5-7%.

#### Summary of the Invention

The glasses of the invention belong to the base system  $SiO_2-B_2O_3-TiO_2-R_2O$  (R = Li, Na, K) and are prepared from a batch composition consisting essentially, in weight percent on the oxide basis:

|    | SiO <sub>2</sub>                      | 49-71 | $\text{Li}_2\text{O} + \text{Na}_2\text{O} + \text{K}_2\text{O}$     | 8-20  |
|----|---------------------------------------|-------|--|-------|
| 10 | B <sub>2</sub> O <sub>3</sub>         | 5-26  | Tio <sub>2</sub> 1   | .8-6  |
|    | 2 3<br>Al <sub>2</sub> 0 <sub>3</sub> | 0-14  | zro <sub>2</sub>   | 0-5.5 |
|    | Li <sub>2</sub> 0                     | 0-4   | As <sub>2</sub> 0 <sub>3</sub> and/or Sb <sub>2</sub> 0 <sub>3</sub> | 0-0.7 |
|    | Na <sub>2</sub> O                     | 0-16  | Cl and/or Br   | 0-1   |
|    | K <sub>2</sub> O                      | 0-20  | CaO and/or MgO   |       |
|    | 2                                     |       |  |       |

and/or ZnO and/or BaO 0-4

The content of SiO<sub>2</sub> lies between about 49-71% approximately and results from the choice of the other glass components whose content limitations are described below.

120 B<sub>2</sub>O<sub>3</sub> is an essential element of the glass because it especially makes it possible to obtain a low density; when B<sub>2</sub>O<sub>3</sub> increases, the density of the glass decreases, the UV cutoff rises and the viscosity of the glass at high temperature decreases, which facilitates its melting and forming. To eliminate B<sub>2</sub>O<sub>3</sub> from the glass would lead, on the one hand, to a density situated at the higher admissible limit, and, on the other hand, to higher melting and forming temperatures. Therefore, the glass will contain at least 5%. Above 26%, the SiO<sub>2</sub> content would have to be greatly reduced,

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thereby reducing the chemical resistance of the glass and the transmission.

The presence of  $Al_2O_3$  is not indispensable; nevertheless, the most preferred glasses will contain some. As a matter of fact, besides improving the chemical resistance,  $Al_2O_3$  substituted for  $SiO_2$  reduces the density and, more significantly, increases the UV cutoff. Above about 14%, the solubility of  $TiO_2$  in the glass is greatly diminished and, in a high content, it precipitates during the course of cooling. If one seeks to obtain glasses having a UV cutoff higher or equal to 325nm, it is indispensable that the  $Al_2O_3$  content be greater than 2-3%, if  $B_2O_3$  is less than about 14%.

TiO, is the determining element for the UV cutoff. 15 Therefore, a minimum amount of 1.8% is necessary to obtain a UV cutoff on the order of 310nm. To attain 325nm, TiO, should be greater than approximately 4% with the previously-stated condition with respect to Al<sub>2</sub>O<sub>3</sub> and B<sub>2</sub>O<sub>3</sub>. Above 6% TiO<sub>2</sub>, the tendency to opacify 20 during the course of cooling is increased and that particularly in the glasses containing a low content of alkali metal oxides (the latter is necessary in order not to exceed the desired index value). Besides its UV absorption, indispensable presence for 25 contributes to the refractive index. Depending upon the content thereof, the index is adjusted with the alkali metal oxides, the concentration of ZrO2, and/or of the alkaline earth metal oxides and/or ZnO.

30 ZrO<sub>2</sub> is preferred over MgO, CaO, BaO, or ZnO for adjusting the index when the alkali metal oxides do not suffice.

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With respect to the alkali metal oxides, the sum  $\text{Li}_2^{O} + \text{Na}_2^{O} + \text{K}_2^{O}$  will range between about 8-20%. Below 8%, the viscosity of the glass is high and the On the other hand, the transmission decreases. solubility of TiO, in the glass decreases which promotes opacification. To avoid phase separation of the matrix, the presence of Al<sub>2</sub>O<sub>3</sub> is recommended when the content of alkali metals is close to the minimum The maximum concentration of alkali metal oxides is determined principally by the density, the 10 refractive index, and the chemical durability. The proportion of the alkali metal oxides is adjusted as a function of the amount of TiO2, B2O3, and Al2O3 in order to obtain the best compromise of properties facility of melting and forming. If utilized alone, Na<sub>2</sub>O and K<sub>2</sub>O will be limited to about 16% and 20%, respectively. Li,0 will not exceed 4% because the tendency of the glass to opalize increases (linked to the insolubility of TiO2). In particular, when TiO2 is greater than 4% in the presence of average amounts of 20 Al203 and B203, the concentration of Li20 will be maintained below 3%. Based upon weight percentages and in substitution with regard to SiO2, the contribution to the index and to the density increases in the following order: K20, Na20, Li20. In general, the UV 25 cutoff diminishes when the quantity of alkali metal oxides increases. Therefore, to obtain a high cutoff (>325nm), the total content will preferably be less than about 15%.

When TiO<sub>2</sub> is greater than 3-3.5%, the glass is found to be more sensitive to the melting conditions in the sense that a coloration ("rose-orange") unacceptable to a glass called "white" can be observed

in certain cases. This is probably linked to the presence of Ti<sup>+3</sup> ions; reducing melting conditions will be avoided. To improve the stability with respect to variations in melting conditions or batch materials, and in order to obtain a "white" glass regardless of the TiO<sub>2</sub> content and particularly at 5% TiO<sub>2</sub>, which leads to a high UV cutoff, As<sub>2</sub>O<sub>3</sub> and/or Sb<sub>2</sub>O<sub>3</sub> will be added. A maximum quantity of 0.7% will be employed. As<sub>2</sub>O<sub>3</sub> is preferred to Sb<sub>2</sub>O<sub>3</sub>.

As 20 and Sb 20 contribute equally to the fining of the glass. In certain cases, those fining agents are not sufficient and chlorine (with or without bromine) will be introduced into the batch materials in the form of chlorine (bromine) compounds such as chloride (bromide) of sodium or potassium. In view of the volatilization during the melting process, only a fraction of the batched quantity will be present in the final glass.

### Prior Art

20 United States Patent No. 2,454,607 discloses glasses designed for sealing to a nickel-iron alloy, the glasses having compositions consisting, in weight percent on the oxide basis, of

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$$\frac{B_2O_3}{2}$$
  $\frac{20-24}{3}$   $\frac{Na_2O}{2}$   $\frac{\ge 2}{2}$   $\frac{A_1O_3}{2}$   $\frac{3-6}{2}$   $\frac{K_2O}{2}$   $\frac{\ge 2}{2}$   $\frac{Li_2O}{2}$   $\frac{7-12}{2}$   $\frac{SiO_2}{2}$   $\frac{56-68}{2}$ 

No mention is made of ophthalmic applications, of a high UV cutoff, or of a colorless glass. Moreover, the  $\operatorname{Li}_2$ O content is much higher than can be tolerated in the subject invention.

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United States Patent No. 4,224,074 describes glass frits suitable for decorating glass, glass-ceramic, and ceramic articles, the frits consisting essentially, in weight percent on the oxide basis, of

| 5 | SiO <sub>2</sub>               | 29-55 | Na <sub>2</sub> O                     | 4-20   |
|---|--------------------------------|-------|---------------------------------------|--------|
|   | B <sub>2</sub> O <sub>3</sub>  | 7-31  | Li <sub>2</sub> 0                     | 0-7    |
|   | Al <sub>2</sub> O <sub>3</sub> | 2-8   | Na <sub>2</sub> 0 + Li <sub>2</sub> 0 | 6-24   |
|   | ZrO_                           | 5-16  | F                                     | 0.75-4 |

No mention is made of ophthalmic applications, of a high UV cutoff, or of a colorless glass. Moreover, fluoride is a required component, which element forms no part of the subject invention, and TiO<sub>2</sub> is merely an optional ingredient.

## Description of Preferred Embodiments

The invention is illustrated through the examples provided in the attached table. The compositions are given in batched weight percentages. materials utilized (and notably the source of SiO2) must be carefully selected in order to obtain a minimum amount of Fe<sub>2</sub>O<sub>3</sub>. As a matter of fact, Fe<sub>2</sub>O<sub>3</sub> in too 20 large a quantity (greater than about 100 PPM in the glass) leads to a yellow coloration. This coloration results from the known interaction between Fe<sub>2</sub>O<sub>3</sub> and TiO2. Particularly for SiO2, one utilizes quartz wherein the Fe<sub>2</sub>O<sub>3</sub> content is on the average less than 10 PPM. The other principal batch materials that can be utilized are boric acid [B(OH)3], boric anhydride, the carbonates of Li<sub>2</sub>O, Na<sub>2</sub>O, and K<sub>2</sub>O, titanium oxide, zirconium oxide, or zircon.

In the presence or not of  $As_2O_3$ , at least 2%  $Na_2O$  or  $K_2O$  will be introduced in the form of nitrates in

order to obtain oxidizing conditions during melting of the mixture of batch materials.

The examples of the invention were prepared from mixtures representing 100 grams to several kilograms of glass. Melting of the batch materials took place at 1250°-1350°C following which the temperature was raised to 1400°-1460°C for homogenizing and fining the glass. Pouring took place after cooling to a temperature allowing either the forming of a bar or the pressing of a disc. The corresponding viscosity ranged between 100-1000 poises. The total time of the operation was on the order of 3-7 hours. After forming, the glass was annealed at about 480°C with a cooling rate to ambient temperature of 60°C/hour; the properties were then determined.

Measurements of refractive index and Abbe number were carried out according to conventional methods (for n<sub>d</sub> the yellow ray of He was used) on glasses having been cooled from 480°C to ambient temperature at the rate of 60°C/hour. Density was measured by the immersion method and expressed in g/cm<sup>3</sup>.

The chemical resistance in acid medium was evaluated by the A.O. test described in the journal Applied Optics, 7, No. 5, page 847, May, 1968. It consists of determining the loss of weight of a polished sample immersed at 25°C for 10 minutes in a 10% by weight aqueous solution of HCl. The loss of weight is expressed in mg/cm<sup>2</sup>.

The transmission curve was recorded at a thickness of 2mm with the aid of a Hewlett-Packard spectrophotometer (type 8450A).

The glasses of the invention can be tinted for an ophthalmic application by using conventional colorants:

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transition metal oxides or rare earth oxides or other known colorant el ments. The total content of these colorants will not generally exceed 2%.

The glasses of the invention can be strengthened according to current techniques of thermal or chemical tempering. In the latter case, the strengthening requires the replacement of a Na<sup>+</sup> and/or Li<sup>+</sup> ion in the glass by Na<sup>+</sup> and K<sup>+</sup> ions in a bath of molten nitrates, which implies that the glass does not contain K<sub>2</sub>O as the only alkali metal oxide.

Preferably, and in order to obtain a UV cutoff higher than about 325nm, the glasses will be prepared from a batch composition consisting essentially, in weight percent on the oxide basis:

| 15 | SiO <sub>2</sub>               | 54-70 | $Li_2O + Na_2O +$ | K <sub>2</sub> O               | 10-13     |
|----|--------------------------------|-------|-------------------|--------------------------------|-----------|
|    | B <sub>2</sub> 0 <sub>3</sub>  | 9-22  | -                 | TiO <sub>2</sub>               | 4-6       |
|    | A1 <sub>2</sub> 0 <sub>3</sub> | 3-10  |                   | ZrO <sub>2</sub>               | 0-2       |
|    | Li <sub>2</sub> O              | 0.5-3 |                   | As <sub>2</sub> 0 <sub>3</sub> | 0-10-0-50 |
|    | Na <sub>2</sub> O              | 3-9   |                   | Cl                             | 0.2-0.7   |
| 20 | к <sub>2</sub> о               | 3-10  | <i>;</i>          | •                              |           |

As previously described, the  ${\rm TiO}_2$  content must be adjusted as a function of the  ${\rm Al}_2{\rm O}_3$ ,  ${\rm B}_2{\rm O}_3$ , and alkalimetal oxide contents in order to obtain a UV cutoff greater or equal to 325nm (see table of examples). For example, a low content of  ${\rm Al}_2{\rm O}_3$  and  ${\rm B}_2{\rm O}_3$  must be associated with a high concentration of  ${\rm TiO}_2$ .

The particularly preferred glasses, because of their excellent properties and their facility for melting and forming on an industrial scale, are

prepared from a batch composition consisting essentially, in weight percent on the oxide basis:

|   | essentially,                   | in weight | pozos                             | w 0                            | 10.5-12.5  |
|---|--------------------------------|-----------|-----------------------------------|--------------------------------|------------|
|   | si <sup>0</sup> 2              | 56-62     | $\text{Li}_2^0 + \text{Na}_2^0 +$ |                                | 4.2-5.5    |
|   | B <sub>2</sub> 03              | 15-20     |                                   | TiO <sub>2</sub>               | 0-1.5      |
| 5 | Al <sub>2</sub> O <sub>3</sub> | 4-7.5     |                                   | ZrO <sub>2</sub>               | 0.10-0.50  |
| - | Li <sub>2</sub> 0              | 1-2.5     |                                   | As <sub>2</sub> O <sub>3</sub> |            |
|   | 2<br>Na <sub>2</sub> 0         | 3-7       |                                   | Cl                             | 0.2-0.7    |
|   | к <sub>2</sub> 0               | 3-6       | forred                            | 1                              | is that of |
|   | ~ ~                            |           |                                   | σ i ass                        | 15         |

The most especially preferred glass is that of

Example 1. This glass has a low density (2.37 g/cm<sup>3</sup>),

a high UV cutoff (330nm), and exhibits no coloration in
the visible. Furthermore, it has excellent chemical
durability (loss of weight in the A.O. test =
0.004mg/cm<sup>2</sup>) and can be easily produced in a continuous

melting unit. It can also be formed at a viscosity of
3000-6000 poises with no problem of devitrification.

#### Table

|    |                                |       |       | 3     | 4     | 5     | 6       |           |  |
|----|--------------------------------|-------|-------|-------|-------|-------|---------|-----------|--|
|    | sio <sub>2</sub>               | 58.89 | 55.91 | 69.21 | 69.21 | 59.26 | 65.69   | 58.89     |  |
|    | B <sub>2</sub> O <sub>3</sub>  | 17.28 | 20.26 | 9.94  | 6.96  | 22.38 | 15.12   | 13.53     |  |
| 5  | Al <sub>2</sub> O <sub>3</sub> | 6.19  | 6.19  | 3.21  | 6.19  | 1.23  | 2.21    | 9.94      |  |
|    | Li <sub>2</sub> O              |       |       |       |       | 1.81  |         |           |  |
|    | Na <sub>2</sub> O              |       | 4.09  |       |       | 6.11  |         |           |  |
| -  | к <sub>2</sub> 0               | 5.81  | 5.81  | 5.81  | 5.81  | 3.84  | 5.79    | 5.81      |  |
|    | TiO <sub>2</sub>               | 5.07  | 4.47  | 5.07  | 5.07  | 4.5   | 4.46    | 4.47      |  |
| 10 | cl                             | 0.57  | 0.57  | 0.57  | 0.57  | 0.57  | 0.57    | 0.57      |  |
|    | As <sub>2</sub> O <sub>3</sub> |       | 0.3   |       |       |       |         |           |  |
|    |                                |       |       |       |       |       | 9 1.522 | 26 1.5227 |  |
|    | n<br>d<br>V                    | 53    |       | 54.1  |       |       | 55.3    |           |  |
|    | V <sub>d</sub><br>D            |       |       |       |       | 2.42  | 2.39    | 2.38      |  |
| 15 | % Trans.                       |       |       |       |       |       |         |           |  |
|    | 400nm                          | 90.1  | 89.7  | 90.1  | 90.4  | 89.9  | 90.2    | 90        |  |
|    | UVnm                           |       |       |       |       |       |         |           |  |
|    | Cutoff                         | 330   | 326   | 325   | 327   | 324   | 323     | 327       |  |

Table - Continued -

|    |                                       | 8     | 9          | 10       | _11      | 12_     | 13_     | 14         |
|----|---------------------------------------|-------|------------|----------|----------|---------|---------|------------|
|    | sio,                                  | 63.58 | 58.9       | 58.91    | 56.33    | 62.13   | 64.35   | 64.35      |
| 5  | B <sub>2</sub> O <sub>3</sub>         | 14.1  | 15.29      |          | 18.15    |         |         |            |
|    | 2 3<br>Al <sub>2</sub> 0 <sub>3</sub> | 3.2   | 6.19       | 6.19     | 6.19     | 6.22    | 3.23    | 3.23       |
|    | Li <sub>2</sub> 0                     | -     | 2.79       | 1.8      | 1.8      | 0.81    | 1.81    | 1.81       |
|    | Na <sub>2</sub> O                     |       |            | 9.89     |          | 6.1     | 4.12    | 4.12       |
|    | к <sub>2</sub> 0                      |       | 8.79       |          |          | 3.83    | 5.85    | 5.85       |
| 10 | MgO                                   | -     | _          | -        | -        | -       | 4       | -          |
|    | CaO                                   | _     | -          | -        | -        | _       | -       | 2          |
|    | ZnO                                   | 1.98  | _          | -        | -        | -       | -       | · <b>-</b> |
|    | ZrO <sub>2</sub>                      | -     | <b>-</b> · | -        | 4.99     | -       | _       | -          |
|    | TiO <sub>2</sub>                      | 4.45  | 4.08       | 4.47     | 2.07     | 5.69    | 3.5     | 3.5        |
| 15 | Cl                                    | 0.56  |            | 0.57     | 0.57     | 0.57    | 0.57    | 0.57       |
|    | As <sub>2</sub> O <sub>3</sub>        | 0.3   | 0.3        | 0.3      | -        |         | 0.3     |            |
|    | n<br>d                                | 1.519 | 1.526      | 59 1.526 | 66 1.520 | 5 1.521 | 2 1.522 | 24 1.5241  |
|    | v<br>đ                                | 54.6  | 55.3       | 54.1     | 57.1     | 51.8    | 57.2    | 56.4       |
|    | d<br>D                                | 2.40  | 2.41       | 2.40     | 2.39     | 2.37    | 2.41    | 2.41       |
| 20 | % Trans.                              |       |            |          |          |         |         |            |
|    | 400nm                                 | 91.2  | 90.6       | 90.7     | 90.7     | 89.5    | 89.4    | 91.8       |
|    | UVnm                                  |       |            |          |          |         |         |            |
|    | Cutoff                                | 323   | 322        | 324      | 311      | 333     | 319     | 320        |

<u>Table</u>
- Continued -

|                                       |                                | 15    | _16     | 17       | _18            | 19       | 20       |           |
|---------------------------------------|--------------------------------|-------|---------|----------|----------------|----------|----------|-----------|
|                                       | SiO <sub>2</sub>               | 64.35 | 58.89   | 57.43    |                | 62.89    |          |           |
| 5                                     | вo                             | 14.27 | 10.55   | 18.15    | 18.07          | 16.16    | 17.87    | 18.18     |
|                                       | Al <sub>2</sub> O <sub>3</sub> | 3.23  | 12.92   | 6.19     | 6.17           | 3.21     | 6.19     | 6.21      |
|                                       | T.i O                          | 1.81  | 2.4     | 1.8      | 1.79           | 1.8      | -        | 0.5       |
|                                       | Na <sub>2</sub> O              | 4.12  | 4.09    | 4.09     | 5.06           | 4.09     | 13.92    | _         |
|                                       | к <sub>2</sub> 0               | 5.85  | 5.81    | 5.81     | 5.78           | 5.81     | -        |           |
| 10                                    | 2<br>ZnO                       | 2     |         |          |                | -        |          | -         |
|                                       | ZrO2                           | _     | _       | 2.98     | 2.97           | 1.99     | -        | ·         |
|                                       | TiO <sub>2</sub>               | 3.5   | 4.47    | 2.98     | 2.97           | 3.48     | 4.47     | 4.48      |
|                                       | C1                             | 0.57  | 0.57    | 0.57     | -              | 0.57     | 0.57     | 0.57      |
|                                       |                                | 0.3   |         |          | -              | -        | 0.3      | 0.3       |
| 15                                    | n <sub>d</sub>                 | 1.520 | 0 1.522 | 28 1.520 | 3 1.525        | 50 1.522 | 21 1.523 | 30 1.5195 |
| · · · · · · · · · · · · · · · · · · · | v <sub>a</sub>                 | 55.8  | 53.7    | 55.8     | 56 <u>.</u> 8_ | 565_     | 54.1     | 54.3      |
|                                       | 'd<br>D                        | 2.40  | 2.39    | 2.38     | 2.42           | 2.39     | 2.42     | 2.38      |
|                                       | % Trans.                       |       |         |          |                |          |          |           |
|                                       | 400nm                          | 90    | 90.5    | 91       | 91.1           | 89.8     | 89.9     | 90.4      |
| 20                                    | UVnm                           |       |         |          |                |          |          |           |
|                                       | Cutoff                         | 320   | 330     | 317      | 316            | 319      | 323      | 323       |

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1. A glass for ophthalmic applications having a refractive index of 1.523 ± 0.004, an Abbe number between 51-59, a density less than 2.43g/cm<sup>3</sup>, a transmission at 400nm in 2mm thickness greater than 89%, and a UV cutoff between 310-335nm, characterized in that it is prepared from a batch composition consisting essentially, in weight percent on the oxide basis, of:

| שנ | 1313, 0-1                      |       | $1.1 O + Na_{0}O + K_{0}O = 8-20$ |
|----|--------------------------------|-------|-----------------------------------|
| 10 | SiO <sub>2</sub>               | 49-71 | 1120 . 112                        |
| 10 | B <sub>2</sub> O <sub>3</sub>  | 5-26  | TiO <sub>2</sub> 1.8-6            |
|    | 2 5                            | 0-14  | zro <sub>2</sub> 0-5.5            |
|    | Al <sub>2</sub> O <sub>3</sub> | 0-4   | $As_2O_3$ and/or $Sb_2O_3$ 0-0.7  |
|    | Li <sub>2</sub> O              | 0-16  | Cl and/or Br 0-1                  |
|    | Na <sub>2</sub> O              | 0-20  | CaO and/or MgO                    |
| 15 | к <sub>2</sub> 0               | 0-20  | and/or ZnO and/or BaO 0-4         |
|    |                                |       |                                   |

2. A glass according to claim 1 having a UV cutoff higher than about 325nm characterized in that it is prepared from a batch composition consisting essentially, in weight percent on the oxide basis, of:

3. A glass according to claim 2 characterized in that it is prepared from a batch composition consisting

essentially, in weight percent on the oxide basis, of:

|   | SiO <sub>2</sub>               | 56-62 | $Li_{2}O + Na_{2}O + K_{2}O$   | 10.5-12.5 |
|---|--------------------------------|-------|--------------------------------|-----------|
|   | B <sub>2</sub> O <sub>3</sub>  | 15-20 | TiO <sub>2</sub>               | 4.2-5.5   |
|   | Al <sub>2</sub> O <sub>3</sub> | 4-7.5 | zro <sub>2</sub>               | 0-1.5     |
| 5 | Li <sub>2</sub> O              | 1-2.5 | As <sub>2</sub> O <sub>3</sub> | 0.10-0.50 |
|   | Na <sub>2</sub> O              | 3-7   | Cl                             | 0.2-0.7   |
|   | к <sub>2</sub> о               | 3-6   |                                |           |

4. A glass according to claim 3 characterized in that it is prepared from a batch composition consisting 10 essentially, in weight percent on the oxide basis, of:

|    | sio,                          | 58.89 | K <sub>2</sub> O               | 5.81   |
|----|-------------------------------|-------|--------------------------------|--|
|    | B <sub>2</sub> O <sub>3</sub> | 17.28 | TiO <sub>2</sub>               | 5.07   |
|    |                               | 6.19  | As <sub>2</sub> 0 <sub>3</sub> | 0.3  |
|    | Li <sub>2</sub> 0             | 1.8   | Cl                             | 0.57   |
| 15 | Na <sub>2</sub> O             | 4.09  |                                |  |
|    | <del>-</del>                  |       |                                | The state of the s |

5. A glass according to claim 1 characterized in that it is tinted by up to 2% by weight relative to the weight of the glass of at least one colorant element.



## EUROPEAN SEARCH REPORT

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| ategory  | of relevant par  | ssages                                    | to class   | <del>"                                    </del> | C 03 C   |        |
| х        | US-A-4 001 019 (T.<br>* Examples; claim *  | YAMASHITA)                                | 1-5  |  | C 03 C<br>C 03 C   | 3/091  |
| x        | BE-A- 551 389 (P)<br>* Page 6, table II  | PG)<br>; claim l *                        | 1-5  |  |  | -4     |
| Х        | CHEMICAL ABSTRACTS 4, 28th January 19 27323u, Columbus, - A - 79 113 618 ( KOGYO K.K.) 05-09- * Abstract *   | Ohio, US; & JP<br>TOSHIBA KASEI           | 1  |  |  |        |
| x        | GB-A-2 115 403<br>(CARL-ZEISS-STIFT)<br>* Examples 1-10 *  | ING)                                      | 1  |  | TECHNICAL<br>SEARCHED  | FIELDS |
|          |  |   |  |  | C 03 C   | 3/00   |
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|          |  | and drawn up for all claims               |  |  |  |        |
|          | The present search report has be   |   | gearch   |  | Examiner<br>RUCHE J.   | P.F.   |
|          | THE HAGUE  | Date of completion of the 20-03-198       |  |  |  |        |
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